

Phase Equilibria in the Fe-Fe₂O₃-Eu₂O₃ System at 1200 °C

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The standard Gibbs energies of the reaction of EuFeO₃ and Eu₃Fe₅O₁₂ were determined to be -63300 ± 200 cal and -297800 ± 600 cal respectively at 1200 °C by means of following reactions: $\text{Fe} + 1/2 \text{Eu}_2\text{O}_3 + 3/4 \text{O}_2 = \text{EuFeO}_3$ and $5 \text{Fe} + 3/2 \text{Eu}_2\text{O}_3 + 15/4 \text{O}_2 = \text{Eu}_3\text{Fe}_5\text{O}_{12}$, based on the phase equilibria in the Fe-Fe₂O₃-Eu₂O₃ system.

Many works on the double oxides involving rare earth sesquioxide and iron oxide have been investigated from the standpoint of crystal chemistry and the magnetic properties of perovskite and garnet structures.¹⁾ Recently, Bedford and Catalano²⁾ have confirmed the existence of EuO, Eu₃O₄, and Eu₂O₃ at 1500 °C; they were first found by Bärnighausen.³⁾ McCarthy and Fischer⁴⁾ have also studied the Eu-Fe-O system at 1200 °C in a flowing argon atmosphere; they have also confirmed these compounds in addition to EuFeO₃ and Eu₃Fe₅O₁₂. However, very few studies have been done on the Gibbs energy of the reaction of lanthanoid-iron -perovskite or -garnet.

Recently, Kimizuka and Katsura⁵⁾ have determined the standard Gibbs energy of the reaction of LaFeO₃ from metallic iron, La₂O₃, and oxygen on the basis of the phase equilibria in the Fe-Fe₂O₃-La₂O₃ system at 1204 °C. They⁶⁾ have also established the phase equilibria in the Fe-Fe₂O₃-Y₂O₃ system at 1200 °C and determined the Gibbs energy of the reactions of YFeO₃, Y₃Fe₅O₁₂, and YFe₂O₄.

In the present study, we have determined the standard Gibbs energy of the reaction of EuFeO₃ and Eu₃Fe₅O₁₂ at 1200 °C on the basis of the phase equilibria in the Fe-Fe₂O₃-Eu₂O₃ system.

Experimental

Materials. High-purity-grade (99.9%) Eu₂O₃ and guaranteed-reagent-grade Fe₂O₃ were employed as the starting substances. Eu₂O₃ was heated at 1200 °C in air for 2 days to obtain B-type Eu₂O₃.⁷⁾ Fe₂O₃ was heated at 900 °C in air for 1 day. The subsequent procedures to obtain the sintered mixtures were the same as those described in a previous paper.⁵⁾

Apparatus and Procedures. The quenching and thermogravimetric methods were adopted in the present study; the details of the procedures have been described by Kimizuka and Katsura.^{5),6)} The weight of the sample in oxygen was chosen as the reference for thermogravimetry. The equilibrium composition was determined by means of two side reactions. The quenching method employed in the present study is the same as that described in a previous paper,⁶⁾ the phases in quenched samples were identified by the powder X-ray diffraction method, with Mn-filtered Fe-K α radiation. The Fe²⁺/Fe³⁺ ratio in the quenched nonstoichiometric EuFeO₃ was determined by the wet chemical method.⁸⁾

Results and Discussion

Phase Equilibria. The equilibrium data at 1200 °C are summarized in Table 1 and are illustrated in Fig. 1. The following phases were stable under the present ex-

TABLE 1. DATA FOR EQUILIBRIA IN THE Fe-Fe₂O₃-Eu₂O₃ SYSTEM AT 1200°C

log P_{O_2} (± 0.02)	Fe	Fe ₂ O ₃ (mole %)	Eu ₂ O ₃	Phases
0.00	0.00	38.00	62.00	E + Pv
	0.00	50.00	50.00	Pv
	0.00	57.00	43.00	Pv + G
	0.00	62.50	37.50	G
	0.00	78.56	21.44	H + G
	0.00	80.00	20.00	H + G
3.88	0.00	90.00	10.00	H + G
	0.00	57.00	43.00	Pv + G
	0.00	62.50	37.50	G
	10.14	70.92	18.94	M + G
	9.36	70.20	20.44	M + G
	15.05	75.70	9.25	M + G
9.14	3.75	53.39	42.91	M + Pv
	6.01	57.61	36.38	M + Pv
	12.50	67.40	20.10	M + Pv
	13.10	68.21	18.69	M + Pv
	16.92	73.93	9.15	M + Pv
	6.42	51.96	41.62	W + Pv
9.27	11.01	53.55	35.44	W + Pv
	23.39	57.68	18.93	W + Pv
	24.26	58.16	17.58	W + Pv
	30.96	60.60	8.44	W + Pv
	9.80	25.44	57.10	W + Pv
	10.26	26.68	55.99	W + Pv
10.82	28.06	54.74	17.20	W + Pv
	11.26	29.31	53.62	W + Pv
	11.92	8.45	50.37	W + Pv
	14.47	50.74	34.79	W + Pv
	29.29	52.42	18.31	W + Pv
	30.65	52.41	16.91	W + Pv
12.09	38.49	53.44	8.07	W + Pv
	23.59	38.48	37.93	Fe + Pv
	40.10	29.92	29.98	Fe + Pv
	72.25	14.05	13.70	Fe + Pv
	75.15	12.36	22.49	Fe + Pv
	88.75	5.69	5.56	Fe + Pv
12.51	0.69	37.53	61.78	E + Pv
	1.58	48.84	49.58	Pv
13.02	76.69	0.19	23.12	Fe + E
	87.61	0.34	12.05	Fe + E
	88.71	0.16	11.13	Fe + E
	94.56	0.17	5.27	Fe + E
15.80	66.66	0.01	33.33	Fe + E
	54.97	0.07	44.96	Fe + E
	72.22	0.31	27.47	Fe + E

Abbreviations have the following meaning: E=Eu₂O₃, Pv=EuFeO₃, G=Eu₃Fe₅O₁₂, H=Fe₂O₃, M=Fe₃O₄, W=FeO, Fe=metallic iron.

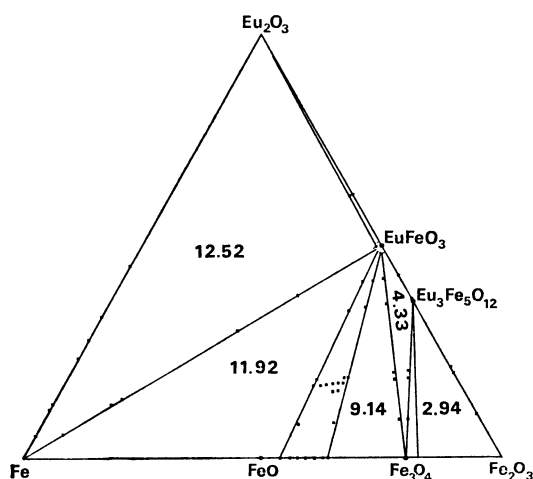


Fig. 1. Phase equilibria in the Fe-Fe₂O₃-Eu₂O₃ system at 1200 °C.

Numbers in the Figure mean values of $-\log P_{O_2}$ at which three crystalline phases are in equilibrium state. Small dots show the experimental results given in Table 1.

perimental conditions: Europium sesquioxide (Eu₂O₃), hematite (Fe₂O₃), magnetite (Fe₃O₄), wüstite (FeO), metallic iron (γ -Fe), europium-iron-perovskite (EuFeO₃), and europium-iron-garnet (Eu₃Fe₅O₁₂). The equilibria among the three condensed phases are as follows: (1) EuFeO₃, Eu₂O₃, and metallic iron, (2) EuFeO₃, wüstite, and metallic iron, (3) EuFeO₃, magnetite, and wüstite, (4) EuFeO₃, Eu₃Fe₅O₁₂, and magnetite, and (5) Eu₃Fe₅O₁₂, magnetite, and hematite. The following phases were characterized with respect to the stoichiometry: (a) Pure Eu₂O₃ was stable down to an oxygen partial pressure of $10^{-15.80}$ atm. However, it showed a small degree of non-stoichiometry at $10^{-15.80}$ atm. McCarthy and White⁹) have reported the approximate phase relation of the Eu-Eu₂O₃ system at elevated temperatures up to 2200 °C; they showed the non-stoichiometry for Eu₂O₃ at 1200 °C and about 10^{-15} atm P_{O_2} , but they did not ascertain the extent of this non-stoichiometry. In the present study, by means of the thermogravimetric method, we determined the composition of the non-stoichiometric Eu₂O₃ phase to be Eu₂O_{2.98±0.02} at $10^{-15.80}$ atm P_{O_2} . The lattice constants of both stoichiometric and non-stoichiometric Eu₂O₃ are given in Table 2. It is supposed that the non-stoichiometric Eu₂O₃ has a larger unit cell volume than the stoichiometric Eu₂O₃; this may be caused by the substitution of Eu²⁺ for Eu³⁺. (b) The deviation from the stoichiometric composition of Eu₃Fe₅O₁₂ was not observed, and the lattice constants of the respective Eu₃Fe₅O₁₂ compounds equilibrated with Fe₂O₃, EuFeO₃ and with Fe₃O₄ were identical with each other. The values of stoichiometric Eu₃Fe₅O₁₂ are identical with those obtained by Espinosa.¹⁰) (c) EuFeO₃ showed a significant deviation from the stoichiometry at relatively low oxygen partial pressures. The degree of non-stoichiometry was determined by both thermogravimetry and wet chemical analysis. These results were consistent with each other. The maximum deviation was

TABLE 2. LATTICE CONSTANTS OF Eu₂O₃ AND EuFeO₃
a) Eu₂O₃ (monoclinic B-type)

Lattice constants	Stoichiometric Eu ₂ O ₃	Non-stoichiometric Eu ₂ O ₃	Hoekstra ¹²⁾
a (Å)	14.102±0.005	14.113±0.004	14.12
b (Å)	3.601±0.001	3.603±0.001	3.600
c (Å)	8.806±0.004	8.808±0.003	8.801
β (deg)	100.02 ±0.3	100.05 ±0.2	99.98
V (Å ³)	440.3 ±0.3	441.0 ±0.2	

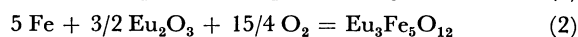
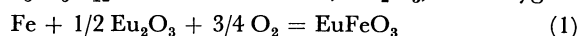
b) EuFeO₃ (Space group: D_{2h}¹⁶-Pbnm, orthorhombic perovskite structure)

Lattice constants	EuFeO ₃ coexisting with Eu ₂ O ₃ Eu ₃ Fe ₅ O ₁₂		Non-stoichiometric EuFeO ₃	Stoichiometric EuFeO ₃	Geller and Wood ¹³⁾
a (Å)	5.373	5.371	5.373	5.373	5.371
b (Å)	5.607	5.605	5.607	5.604	5.611
c (Å)	7.683	7.684	7.685	7.684	7.686
V (Å ³)	231.4	231.3	231.5	231.3	231.6

Deviations in *a*, *b*, and *c* are ±0.001 by the least squares method, and those of *V* ±0.1.

determined to be EuFeO_{2.976} at an oxygen partial pressure of $10^{-12.52}$ atm; at this point the non-stoichiometric EuFeO₃ was in equilibrium with both metallic iron and stoichiometric Eu₂O₃. EuFeO₃ shows a small but equivocal solubility of the three components. Since we could not determine the exact boundary of the EuFeO₃ field, the EuFeO₃ field was approximately bounded by the dotted line shown in Fig. 1. The lattice constants of EuFeO₃ in equilibrium with Eu₂O₃ and Eu₃Fe₅O₁₂, and of non-stoichiometric EuFeO₃ with the composition of EuFeO_{2.976}, were compared with those of stoichiometric EuFeO₃ given in Table 2.

(2) Calculation of the Standard Gibbs Energy of the Reaction of EuFeO₃ and Eu₃Fe₅O₁₂. On the basis of the phase equilibria, we may determine the following two standard Gibbs energies of the reaction of EuFeO₃ and Eu₃Fe₅O₁₂ from metallic iron, Eu₂O₃, and oxygen:

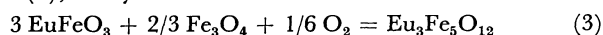


Since the EuFeO₃ component in equilibrium with metallic iron and Eu₂O₃ has the composition of EuFeO_{2.976}, it is convenient to choose the reference activity of the EuFeO₃ component as unity at this composition. As has been described before, the Eu₂O₃ component had a small compositional variation at lower oxygen partial pressures. However, the composition of Eu₂O₃ was nearly stoichiometric at an oxygen partial pressure of $10^{-12.52}$ atm, at which Reaction (1) was in the equilibrium state. Thus, with reference to Eq. (1), we may determine the standard Gibbs energy of the reaction of EuFeO₃ as follows

$$\Delta G^0(1) = 3/4 RT \ln P_{O_2} = -63300 \pm 200 \text{ cal} \cdot \text{mol}^{-1},$$

where *R* and *T* indicate the gas constant and the absolute temperature respectively. The experimental error was estimated on the basis of the fluctuations of both temperature and the oxygen partial pressure.

Because the stoichiometric $\text{Eu}_3\text{Fe}_5\text{O}_{12}$ was in equilibrium with both stoichiometric EuFeO_3 and Fe_3O_4 at an oxygen partial pressure of $10^{-4.33}$ atm, the standard Gibbs energy change, referred to the following equation (3), may be calculated to be -4900 ± 200 cal:



According to Darken and Gurry,¹¹⁾ the activity of metallic iron in stoichiometric magnetite in equilibrium with wüstite is $10^{-1.545}$ at 1200 °C, while the oxygen partial pressure related to the wüstite-magnetite equilibrium is $10^{-9.14}$ atm. Thus, the standard Gibbs energy of the formation of Fe_3O_4 , $\Delta G^0(4)$, referred to the following equation (4) may be calculated to be -154500 ± 300 cal at 1200 °C:



By adding $\Delta G^0(3)$, $2/3\Delta G^0(4)$, and $3\Delta G^0(1)$, we obtain the standard Gibbs energy of the reaction of $\text{Eu}_3\text{Fe}_5\text{O}_{12}$, $\Delta G^0(2)$, referred to Equation (2), as -297800 ± 600 cal.

It is interesting to compare the present Gibbs energy data with those of LaFeO_3 , YFeO_3 , and $\text{Y}_3\text{Fe}_5\text{O}_{12}$ which have been determined by Kimizuka and Katsura.^{5,6)} Comparing these data, we may observe that the standard Gibbs energy of the reaction of lanthanoid-iron perovskite from metallic iron, lanthanoid sesquioxide, and oxygen increases with decrease in the ionic radii of lanthanoid ions. The relationship between the ionic radii and the Gibbs energy will be discussed in more detail in the near future.

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References

- 1) G. Brauer, in "Progress in the Science and Technology of the Rare Earths," (L. Eyring, Ed.), Vol. 1, pp. 152—166, Pergamon Press, New York (1964); R. S. Roth, *ibid.*, pp. 167—210; E. F. Westrum, *ibid.*, pp. 310—350; N. A. Toropov, V. P. Barzakovskii, V. V. Lapin, and N. N. Kurtseva, in "Handbook of Phase Diagrams of Silicate Systems," Vol. 1, pp. 603—610, translated from Russian, Israel Program for Scientific Translations, Israel (1972); R. S. Roth, *J. Res. Natl. Bur. Stand.*, **58**, No. 2, 75 (1957); R. S. Roth and S. J. Schneider, *J. Res. Nat. Bur. Stand.*, **64A**, No. 4, 309 (1960); S. J. Schneider, R. S. Roth, and J. L. Waring, *ibid.*, **65A**, No. 4, 345 (1961).
- 2) R. G. Bedford and E. Catalanolo, *J. Solid State Chem.*, **3**, 112 (1971).
- 3) H. Bärnighausen, *J. Prakt. Chem.*, **34**, 1 (1965).
- 4) G. J. McCarthy and R. D. Fischer, *J. Solid State Chem.*, **4**, 340 (1972).
- 5) N. Kimizuka and T. Katsura, This Bulletin, **47**, 1801 (1974).
- 6) N. Kimizuka and Katsura, *J. Solid State Chem.*, **13**, 176 (1975). Standard Free Energy of Formation of YFeO_3 , $\text{Y}_3\text{Fe}_5\text{O}_{12}$, and a New Compound YFe_2O_4 in the $\text{Fe}-\text{Fe}_2\text{O}_3-\text{Y}_2\text{O}_3$ system at 1200 °C.
- 7) R. S. Roth and S. J. Schneider, *J. Res. Nat. Bur. Stand.*, **64A**, 309 (1960).
- 8) I. Iwasaki, T. Katsura, M. Yoshida, and T. Tarutani, *Bunseki Kagaku*, **6**, 211 (1957).
- 9) G. J. McCarthy and W. B. White, *J. Less Common Metals*, **22**, 409 (1970).
- 10) G. P. Espinosa, *J. Chem. Phys.*, **37**, 2344 (1962).
- 11) L. S. Darken and R. W. Gurry, *J. Am. Chem. Soc.*, **67**, 1398 (1945).
- 12) H. R. Hoekstra, *Inorg. Chem.*, **5**, 754 (1966).
- 13) S. Geller and E. A. Wood, *Acta Crystallogr.*, **9**, 563 (1956).